Contents lists available at ScienceDirect



Materials Science in Semiconductor Processing

journal homepage: www.elsevier.com/locate/mssp

Optimization of electroless plating of gold during MACE for through etching of silicon wafer



Muni Raj Maurya^{a,c}, Vijaykumar Toutam^{a,*}, Preetam Singh^a, Sivaiah Bathula^b

^a CSIR-National Physical Laboratory, Dr. K. S. Krishnan Marg, New Delhi, 110012, India

^b Advanced Materials and Devices Metrology Division, CSIR-National Physical Laboratory, Dr. K. S. Krishnan Marg, New Delhi, 110012, India

^c Academy of Scientific and Innovative Research (AcSIR), CSIR-National Physical Laboratory, Dr. K. S. Krishnan Marg, New Delhi, 110012, India

A R	ΤI	CLE	INF	0

Keywords: Metal assisted chemical etching Gold Electroless plating Silicon Through etching

ABSTRACT

Deep etching of silicon (Si) is very much desirable for wide variety of applications. Under the context, a cost effective and reproducible through etching of \sim 375 µm thick Si wafer is demonstrated through long hour metal assisted chemical etching (MACE) followed by short duration KOH etching. During MACE, apart from pH and temperature, metal catalyst size and coverage density during electroless plating plays an important role. Optimization of gold deposition in terms of plating solution concentration and deposition time during MACE is studied for effective through etching. HAuCl₄ concentration of \sim 5 mM for 30 s is found to be best suited for MACE and produces deep and highly dense pores in Si with threshold pore radius \sim 250 nm and above. Following the MACE, KOH etching effectively scoops out porous Si to realize through etching.

1. Introduction

Silicon (Si) micromachining is extensively used for the fabrication of patterns and complex three dimensional structures. Introduction of new techniques like surface micromachining [1], anisotropic etching [2], LIGA process [3] and Bosch process [4] has revolutionized its application in microfluidics [5] and micro-electro-mechanical systems (MEMS) devices [6] etc. Deep etching of Si by dry and wet anisotropic etching requires a hard mask of Si₃N₄/SiO₂. Many of the techniques and the processes like low pressure chemical vapor deposition (LPCVD), plasma-enhanced chemical vapor deposition (PECVD) and thermal oxidation are carried out in controlled environment where pressure, flow rate of gases, temperature etc needs to be maintained for high quality growth of Si₃N₄/SiO₂ [7]. Considering these many variable parameters in the growth process and the maintenance of instrument increases the cost of production. Thus, a simple wet etch techniques using polymer as a mask will be viable and useful for many researchers to practice without the need of such high end facilities. Silicon anisotropic wet etching is done in hydroxide containing solution, where crystal planes with different hybridised SP³ orbital effects the etch rate drastically [8]. Isotropic wet etching of Si in HF solution is well known which happens in the presence of holes (h^+) , produced by oxidative chemical reaction or supplied externally under bias [9]. Etch profile can be restricted to certain region by trapping hole concentration in selected geometry. HF based metal assisted chemical etching (MACE)

method was reported by Li and Bohn [10]. Metals like gold (Au), silver, platinum and copper are patterned on Si in the desired region of etching [11]. Due to catalytic activity of metal, concentration of holes in the metal vicinity is more, which intern increases Si rate of etching in these regions. MACE has received much attention in recent years due to its low cost and control over etched morphology. Chartier et al. used molar concentration ratio (ρ) = [HF] / ([HF] + [H₂O₂]) to characterize different etched morphologies [12]. Effect of catalyst shape and concentration, on etched direction was studied by Hildreth et al. [13]. Controlled 3D motion of insulator pinned metal catalyst was also demonstrated by Hildreth et al. [14]. Silicon micromachining using MACE produces, porous Si, Si nanowires, patterned structure and have application in trench capacitor, antireflective surface, thermal conversion, and bio-mimic super-hydrophobicity etc [15,16].

Deep etching of Si by wet method requires hard mask or metal as catalyst. Fabrication of through wafer micro funnel has been reported previously [17]. Dry etching in combination with anisotropic KOH etching is adopted for realization of high density devices and interconnects in packaging. Recently, MACE based deep etching of Si is reported, where metal is deposited by evaporation or sputtering technique [18]. Implementation of these processes in large-scale manufacturing increase the production cost. Thus, a simpler and a cost effective method entirely based on wet synthesis is required for deep or through etching of Si wafer, resulting in large scale production of devices.

* Corresponding author.

E-mail address: toutamvk@nplindia.org (V. Toutam).

https://doi.org/10.1016/j.mssp.2019.04.046

Received 12 March 2019; Received in revised form 17 April 2019; Accepted 29 April 2019 Available online 11 May 2019

1369-8001/ © 2019 Elsevier Ltd. All rights reserved.

In the present work we report through etching of Si wafer using electroless plated Au as catalyst and polymer as mask. Long hour MACE is adopted to make deep and high density pores in Si, followed by short duration KOH etching to scoop out porous Si and reveal effective etch depth. Gold electroless plating concentration and deposition time is optimized to get highly porous Si during MACE having threshold pore radius and good coverage. The fabricated through hole in Si wafer can be used as window in analysis of optical detector with out of plane contacts [19], through wafer interconnects and is best suited for high density device fabrication.

2. Experiment

Boron doped P-type Si wafer < 100 > with resistivity 1-10 Ω -cm and thickness \sim 375 µm were used in experiments. Polished side of the wafer has SiO₂ layer of thickness \sim 200 nm. The wafer was cut in to $10 \times 10 \text{ mm}^2$ pieces. Samples were cleaned by sonicating in acetone for 10 min followed by degreasing in ethanol and finally washing with IP and DI water. Samples were functionalized for hydrogen terminated surface by immersing in DI water: HF (100:1 v/v) solution for 1 min. Prebaking of samples were done at 110 °C for 2 min. PMMA (MW 996,000) solution of 0.35 mM in CHCl₃ is used for spin coating at 3000 rpm for 90 s and samples were baked at 90 °C for 60 min in an oven. A window of size $1 \times 4 \text{ mm}^2$ was created in PMMA through exposure technique and SiO₂ in the exposed region was etched using HF solution. Electroless plating of Au was done in the open window of Si surface for different duration of 5, 10, 20, 30 s, and for different concentrations of 0.1, 1, 5 and 10 mM of HAuCl₄.3H₂O in 5 M HF. For easy understanding, all samples treated for Au deposition in various Au salt concentrations are denoted as sample A, B, C and D for 0.1, 1, 5 and 10 mM, respectively and sample (x, y) notation is used, where y stands for deposition time of 5, 10, 20 and 30 s. For etch rate dependence and etch depth analysis based on Au coverage, MACE of all samples was done for 5 h in a solution of 2.6 M HF and 8.1 M H₂O₂ having molar concentration ratio (ρ) ~0.24, followed by KOH etching in 30 wt% of solution at 60 °C for 15 min. For through etching of Si wafer, a 30 h prolonged MACE was done followed by KOH etching for 15 min.

Field emission scanning electron microscope (Zeiss) was used for the morphological observations of the samples. Stylus profiler (AMBIOS XP-200) is used to analyse the etch depth of samples.

3. Results and discussion

Fig. 1 shows SEM image of Au deposited on Si by electroless plating for 30 s in various concentration of 0.1, 1, 5 & 10 mM Au precursor. For sample (A, 30), clusters of Au nanoparticles with very low coverage density are formed, as shown in Fig. 1a. The inset of the figure shows initial growth of nanoparticle and seed growth during the cluster formation. With increase in metal precursor concentration for sample (B, 30), agglomeration of clusters takes place forming Au islands, and metal coverage density on substrate increases. The inset of sample (B, 30) shows the growth of highly porous interconnecting network, tending to form film. For sample (C, 30) and sample (D, 30), formation of Au film is observed. Inset of respective figures shows that the film formed in sample (C, 30) is more porous with wider cracks as compared to film formed in sample (D, 30).

Fig. 2 shows SEM image of porous Si wafer after 5 h MACE for Au plated Si under various molar concentrations and deposition time of 30 s, along with pore radius distribution. Centre figure summarizes the behaviour of mean pore radius and density of pores for the threshold radius (r) ~ 250 nm and above. For sample (A, 30), MACE produces corrugated Si surface with pores having mean radius ~ 140 nm, as shown in Fig. 2a. With increase in Au plating concentration, for sample (B, 30) size of deposited Au clusters increases forming a network and increase in coverage density (see Fig. 1b). As a result, wider pores are formed with mean pore radius ~ 200 nm, as shown in Fig. 2b. Similar

trend is observed for sample (C, 30). Complete Si surface is covered with bigger clusters during electroless plating forming a porous film (see Fig. 1c). MACE of sample (C, 30) leads to almost regular size pores with mean pore radius $\,\sim\!250\,\text{nm}$ and good coverage density, as shown in Fig. 2c. With further increase in Au plating concentration, for sample (D, 30), shallow etch pits are formed with very few and wide deep pores. Radius distribution of shallow etch pits is shown in the inset of Fig. 2d and has a peak at \sim 400 nm. For compact and thicker Au films, etch rate during MACE decreases as the chemical species have to diffuse over large distance [20]. As a result, uniform etching with shallow etch pits form compared to porous and deep etching observed for porous metal films. For porous Au film with increase in cluster size, pore size, etch depth increases due to enhanced transport of chemical species and redox reaction within the pore [21]. Even though SEM image won't give quantitative information about the depth of the pores, from the contrast of the pores it can be emphasized qualitatively that the depth of the pores is maximum for sample (C, 30). Thus, pores having threshold pore radius ~250 nm and above with high coverage density contributes to highly porous structure and are suitable for through etching of Si wafer. Centre plot in Fig. 2 summarizes the mean pore radius and density of pores with threshold radius ~ 250 nm and above for different Au plated Si wafers. Left y-axis of the plot shows the variation of mean pore radius with increase in Au plating concentration. Clearly as the plating concentration increase the mean pore radius increases due to increase in particle size of catalyst. Right y-axis shows the variation of pore density with Au plating concentration for threshold pore radius \sim 250 nm and above. From the plot it is observed that sample (C, 30) has highest pore density above threshold pore radius, and is best suited for through etching of Si wafer.

Fig. 3 shows change in morphology of single MACE etch pore after short duration of KOH etching and their evolution by merging with adjacent pores forming a larger pore. MACE of sample (C, 30) leads to formation of pores above threshold radius. The short KOH treatment widens the pore and will terminates at < 111 > facet making an angle ~54.7° with < 100 > plane [7]. The lateral widening (l_{shift}) of the pore by KOH etching for time (t) min can be calculated using the formula given in Fig. 3a. The etch rates of the Si < 111 > plane (R < 111 >) in 30 wt% KOH at 70 °C is ~5 nm/min [22]. For 15 min of KOH etching the calculated lateral shift is ~90 nm. For MACE etched pores having inter-pore distance ~180 nm, pore walls collapse after KOH treatment and they merge to form a bigger pore, as shown in Fig. 3b. Pores formed after KOH etching has straight edges due to anisotropic etching of Si.

Fig. 4 shows SEM image of sample (C, 30) after 5 h MACE followed by 10-15 min KOH etching along with stylus plot. Fig. 4a shows SEM image of sample (C, 30) after 5 h MACE with two contrast regions of 10 and 15 min KOH etching. From Fig. 4a, it is observed that the left half of the sample surface after 10 min KOH etching retains the porous structure of Si. Whereas, the right half of the sample after 15 min KOH treatment leaves a flat Si surface, scooping out the whole porous structure and the effective MACE etch depth is exposed. Fig. 4b shows cross sectional SEM image of sample (C, 30) after 5 h MACE and 10 min KOH etching. A gradient in the etch profile is observed with maximum etch depth at the centre of the etched window. This is due to non uniform deposition of the Au at the edge of the window covered with polymer, resulting in non uniform etch depth during MACE. At the centre region of the window where Au plating is uniform, maximum etch depth ~ $60 \,\mu m$ is observed. The leftover Si morphology seen in the etched window is due to 10 min KOH etching and can be removed after extended 5 min KOH etching. The magnified SEM image of the Si morphology at the centre of etched window after 5 h MACE and 10 min of KOH etching is shown in Fig. 4c. Silicon pillars of $\sim 10 \,\mu m$ length are observed. Silicon pillars are mostly round in shape with tapered top, resulting in formation of cones, as shown in the inset Fig. 4c. The evolution of cone shape morphology is attributed to the anisotropic etching property of the KOH. Fig. 4d shows line profile analysis of etch depth after 5 h MACE followed by 10 min KOH etching of sample (C,



Fig. 1. SEM images of electroless plated Au on Si substrate for 30 s deposition time in various molar concentration of (a) 0.1 mM. (b) 1 mM. (c) 5 mM. (d) 10 mM, respectively.



Fig. 2. SEM images of Si wafer after 5 h MACE, using 30 s electroless plated Au in various molar concentrations as catalyst, along with pore radius distribution in the inset. (a) Sample (A, 30). (b) Sample (B, 30). (c) Sample (C, 30). (d) Sample (D, 30). Centre figure shows mean pore radius and density of pores with threshold radius (r) ~ 250 nm and above, for Au plated Si in different concentrations and deposition time of 30 s after 5 h MACE.



Fig. 3. Morphology evolution of MACE etched pore after short KOH treatment. (a) Schematic showing change in pore morphology after KOH etching. (b) SEM image of the pore having almost square periphery after short duration of KOH treatment.

30). A U-shaped etch profile with vertical depth of $54 \pm 3 \,\mu\text{m}$ having slant edges is observed, as shown in Fig. 4d. The fluctuation in the height at the base of the etched pit is due to presence of Si pillars after 10 min KOH etching.

Fig. 5 shows etch depth analysis after 5 h MACE followed by 15 min KOH etching of Au plated Si in various molar concentration and deposition time along with array of through holes in Si. For sample A, etch depth of less than $\sim 5 \,\mu m$ and no dependence on Au deposition time is observed, as shown in Fig. 5a (red curve). In sample A, isolated clusters of gold are deposited for various electroless plating time. MACE of these samples leads to formation of widely separated pores in the Si. Thus, KOH etching only widens the pores and leads to facet termination. As a result, effective MACE etch depth is not exposed. Fig. 5a (black curve) shows the etch depth of sample B for various deposition time. For sample (B, 5), clusters of Au nanoparticles are deposited and the density of these clusters increases in sample (B, 10). The MACE + KOH etch depths of sample (B, 5) and (B, 10) are almost similar up to $17 \pm 2 \mu m$. With increase in deposition time for sample (B, 20) and sample (B, 30), interconnected islands of Au are deposited on the Si surface. As a result, pore radius and coverage area increases during MACE. Thus, surface exposed for KOH etching is more and it scoops out porous Si to reveal effective MACE etch depth of 43 \pm 3 μ m. For sample C, irrespective of deposition time porous Au film is deposited on the Si substrate. With

further increase in aggregation of Au in sample (C, 30), MACE etch depth increases and leads to high density of wide pores. As a result, KOH scoops out porous Si morphology and actual MACE etch depth of 54 \pm 3 µm is exposed, as shown in Fig. 5a (cyan curve). Compared to etch depth of sample C, for sample D there is no dependence of etch depth on Au deposition time and fluctuates about mean etch depth of $45 \pm 11 \,\mu\text{m}$, as shown in Fig. 5a (blue curve). As the concentration of Au salt increases beyond 5 mM, compactness of the Au film increases and MACE etching is more at the edges due to ease in mass transfer. Due to imbalance in spatial etching, cracks are induced in the film leading to fluctuating etch rate and won't show any systematic dependence on Au plating time. Thus, gold deposition in molar concentration ~5 mM and deposition time ~ 30 s is best suited for through etching of Si wafer. Fig. 5b shows array of through holes in \sim 375 µm thick Si wafer by long hour MACE followed by short KOH etching. Array of 3×3 window on Si with area of $1 \times 1 \text{ mm}^2$ were fabricated using exposure technique with PMMA as protective layer and Au was deposited in 5 mM Au salt solution for 30 s. MACE of 30 h followed by 15 min KOH etching results in formation of array of through holes in \sim 375 µm thick Si wafer as shown in Fig. 5b. Non-uniformity in the shape of through etched holes can be overcome by having abrupt polymer edges in the etched window. The proposed deep etching of Si can be used in through wafer interconnects for high density device fabrication and as window in



Fig. 4. SEM images of sample (C, 30) after 5 h MACE followed by short KOH etching along with stylus plot. (a) SEM image after 10, 15 min KOH etching. (b, c) Cross sectional, zoomed in SEM image after 10 min KOH etching, respectively. (d) Line profile analysis of etch depth after 5 h MACE and 10 min KOH etching.



Fig. 5. MACE + KOH etch depth analysis using stylus profiler for Au plating in various molar concentration and deposition time, along with array of through holes in Si wafer. (a) Etch depth analysis after 5 h MACE followed by 15 min KOH etching. (b) Array of 3×3 through holes in ~375 µm thick Si wafer after 30 h MACE using 5 mM and 30 s electroless plated Au as catalyst followed by 15 min of KOH etching.

analysis of optical detector with out of plane contacts [19]. Further, the work is focused on the effect of temperature and inclusion of more electronegative metal like silver to enhance the MACE rate of etching.

4. Conclusion

Deep etching of Si wafer is studied with combination of MACE and KOH etching. Electroless plating of Au in various molar concentrations is adopted for MACE etching of Si wafer. Etch depth is analysed in terms of metal coverage density using stylus profiler. High metal coverage density, results in formation of deep, wide and high density pores in Si wafer. Gold deposition in molar concentration ~ 5 mM and deposition time ~ 30 s is best suited for through etching of Si wafer.

Acknowledgements

Mr. Muni Raj acknowledges Council of Scientific and Industrial Research (CSIR) & Academy of Scientific and Innovative Research (AcSIR) for fellowship and academic support. Dr. Toutam acknowledges DST-SERB for their financial support under FTYS project (GAP -161232).

References

- J.M. Bustillo, R.T. Howe, R.S. Muller, Surface micromachining for microelectromechanical systems, Proc. IEEE 86 (1998) 1552–1574 https://ieeexplore.ieee.org/ abstract/document/704260.
- [2] K.E. Bean, Anisotropic etching of silicon, IEEE Trans. Electron Devices 25 (1978) 1185–1193 https://ieeexplore.ieee.org/stamp/stamp.jsp?tp = &arnumber = 1479644&tag = 1.
- [3] E.W. Becker, W. Ehrfeld, P. Hagmann, A. Maner, D. Münchmeyer, Fabrication of microstructures with high aspect ratios and great structural heights by synchrotron radiation lithography, galvanoforming, and plastic moulding (LIGA process), Microelectron. Eng. 4 (1986) 35–56 https://doi.org/10.1016/0167-9317(86) 90004-3.
- [4] F. Laermer, A. Schilp, Method of anisotropically etching silicon, U. S. Patent No. 5501893 (3, 26, 1996). https://patentimages.storage.googleapis.com/69/2d/c4/ af6ae016bc71da/US5501893.pdf.
- [5] M.J.D. Boer, R.W. Tjerkstra, J.W. Berenschot, H.V. Jansen, G.J. Burger, J.G.E. Gardeniers, M. Elwenspoek, A.V.D. Berg, Micromachining of buried micro channels in silicon, J. Microelectromech. Syst. 9 (2000) 94–103 https://ieeexplore. ieee.org/document/825783.
- [6] J.W. Judy, Microelectro-mechanical systems (MEMS): fabrication, design and applications, Smart Mater. Struct. 10 (2001) 1115–1134 https://doi.org/10.1088/ 0964-1726/10/6/301.

- [7] G.T.A. Kovacs, N.I. Maluf, K.E. Petersen, Bulk micromachining of silicon, Proc. IEEE 86 (1998) 1536–1551 https://ieeexplore.ieee.org/document/704259.
- [8] W. Lang, Silicon microstructuring technology, Mater. Sci. Eng. R Rep. 17 (1996) 1–55 https://doi.org/10.1016/0927-796X(96)00190-8.
- [9] V. Lehmann, H. Föll, Formation mechanism and properties of electrochemically etched trenches in n-type silicon, J. Electrochem. Soc. 137 (1990) 653–659 http:// ies.ecsdl.org/content/137/2/653.
- [10] X. Li, P.W. Bohn, Metal-assisted chemical etching in HF/H₂O₂ produces porous silicon, Appl. Phys. Lett. 77 (2000) 2572–2574 https://aip.scitation.org/doi/10. 1063/1.1319191.
- [11] T. Fatima, B.M. Jeffrey, M.D. Lauren, N. Logan, D. Wenqi, P.J. Michael, Y. Joanne, F. Joanne, R.B. Marcie, Nanostructured silicon via metal assisted catalyzed etch (MaCE): chemistry fundamentals and pattern engineering, Nanotechnology 27 (2016) 412003 https://doi.org/10.1088/0957-4484/27/41/412003.
- [12] C. Chartier, S. Bastide, C. Lévy-Clément, Metal-assisted chemical etching of silicon in HF-H₂O₂, Electrochim. Acta 53 (2008) 5509–5516 https://doi.org/10.1016/j. electacta.2008.03.009.
- [13] O.J. Hildreth, W. Lin, C.P. Wong, Effect of catalyst shape and etchant composition on etching direction in metal-assisted chemical etching of silicon to fabricate 3D nanostructures, ACS Nano 3 (2009) 4033–4042 https://pubs.acs.org/doi/abs/10. 1021/nn901174e.
- [14] O.J. Hildreth, D. Brown, C.P. Wong, 3D out-of-Plane rotational etching with pinned catalysts in metal-assisted chemical etching of silicon, Adv. Funct. Mater. 21 (2011) 3119–3128 https://doi.org/10.1002/adfm.201100279.
- [15] H. Han, Z. Huang, W. Lee, Metal-assisted chemical etching of silicon and nanotechnology applications, Nano Today 9 (2014) 271–304 https://doi.org/10.1016/j. nantod.2014.04.013.
- [16] Y. Xiu, L. Zhu, D.W. Hess, C.P. Wong, Hierarchical silicon etched structures for controlled hydrophobicity/superhydrophobicity, Nano Lett. 7 (2007) 3388–3393 https://pubs.acs.org/doi/10.1021/nl0717457.
- [17] J.M. Archer, S.F. Ligler, Fabrication and characterization of silicon micro-funnels and tapered micro-channels for stochastic sensing applications, Sensors 8 (2008) 3848–3872 https://doi.org/10.3390/s8063848.
- [18] B. Miao, J. Zhang, X. Ding, D. Wu, Y. Wu, W. Lu, J. Li, Improved metal assisted chemical etching method for uniform, vertical and deep silicon structure, J. Micromech. Microeng. 27 (2017) 055019https://doi.org/10.1088/1361-6439/ aa6872.
- [19] M.R. Maurya, V. Toutam, Fast response UV detection based on waveguide characteristics of vertically grown ZnO nanorods partially embedded in anodic alumina template, Nanotechnology 30 (2018) 085704 https://doi.org/10.1088/1361-6528/ aaf545.
- [20] H.D. Um, N. Kim, K. Lee, I. Hwang, J. Hoon Seo, Y.J. Yu, P. Duane, M. Wober, K. Seo, Versatile control of metal-assisted chemical etching for vertical silicon microwire arrays and their photovoltaic applications, Sci. Rep. 5 (2015) 11277 https://doi.org/10.1038/srep11277.
- [21] C.L. Lee, K. Tsujino, Y. Kanda, S. Ikeda, M. Matsumura, Pore formation in silicon by wet etching using micrometre-sized metal particles as catalysts, J. Mater. Chem. 18 (2008) 1015–1020 https://doi.org/10.1039/B715639A.
- [22] K. Sato, M. Shikida, Y. Matsushima, T. Yamashiro, K. Asaumi, Y. Iriye, M. Yamamoto, Characterization of orientation-dependent etching properties of single-crystal silicon: effects of KOH concentration, Sens. Actuators, A-Phys. 64 (1998) 87–93 https://doi.org/10.1016/S0924-4247(97)01658-0.